

# HYDROTHERMAL SYNTHESIS OF $\text{Co}_3\text{O}_4$ NANO-OCTAHEDRA AND THEIR MAGNETIC PROPERTIES

Ana Fernández-Osorio<sup>1</sup>, América Vázquez-Olmos<sup>2</sup>, Roberto Sato-Berru<sup>2</sup> and Roberto Escudero<sup>3</sup>

<sup>1</sup>Facultad de Estudios Superiores Cuautitlán, Universidad Nacional Autónoma de México, A. P. 25, 54740 Edo de Méx., México

<sup>2</sup>Centro de Ciencias Aplicadas y Desarrollo Tecnológico, Universidad Nacional Autónoma de México, A. P. 70-186, 04510, México D.F. México

<sup>3</sup>Instituto de Investigaciones en Materiales, Universidad Nacional Autónoma de México. A.P.70-360, 04510, México D.F. México

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**Abstract.** Highly uniform cobalt oxide ( $\text{Co}_3\text{O}_4$ ) nano-octahedra with mean edge length about  $16.4 \pm 3.1$  nm have been prepared using a hydrothermal method. X-ray diffraction pattern shows the normal spinel structure with formula  $\text{Co}^{2+}(\text{Co}^{3+})_2\text{O}_4$  as the only crystallographic phase. The  $\text{Co}_3\text{O}_4$  nanoparticles were characterized by UV-Vis and Raman spectroscopies and its morphology was determined by scanning and high resolution transmission electron microscopies. Magnetic properties of  $\text{Co}_3\text{O}_4$  nano-octahedra were determined with a MPMS SQUID magnetometer. The blocking temperature ( $T_b$ ) at 8K and a slight hysteresis loop indicating a ferrimagnetic behavior were observed. The magnetic response could be explained by uncompensated surface spins of the  $\text{Co}_3\text{O}_4$  nanoparticles.

## 1. INTRODUCTION

Cobalt oxide ( $\text{Co}_3\text{O}_4$ ) is a promising material for use as a gas sensor and catalyst in hydrocracking processes of crude fuels, pigment for glasses and ceramic. [1-5]. Highly dispersed nanostructured spinel cobalt oxide is expected to display better performance in the above mentioned application aspects. Specific morphologies and crystallographic phases of nanostructures materials are responsible for their optical, magnetic and electric properties [6].

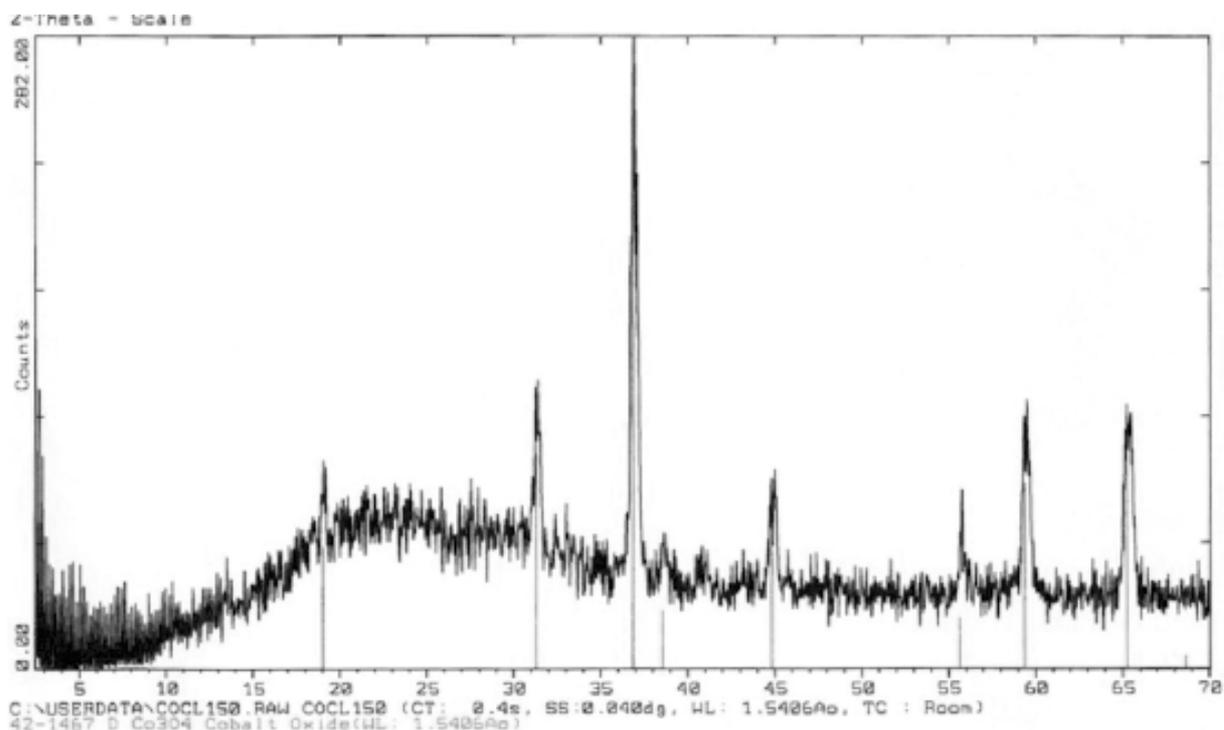
$\text{Co}_3\text{O}_4$  belongs to the normal spinel structure, which is based on a cubic close packing array of oxide ions in which Co(II) ions occupy the tetrahedral 8a sites and Co(III) ions occupy the octahedral 16d sites [7]. Synthesis of cobalt oxide nanoparticles have been obtained by different

methods as solvothermal, mechanochemical, reduction–oxidation, sol-gel and polymer combustion, generating different morphologies like nanotubes, nanorods, nanocubes, and spherical particles [8-19].

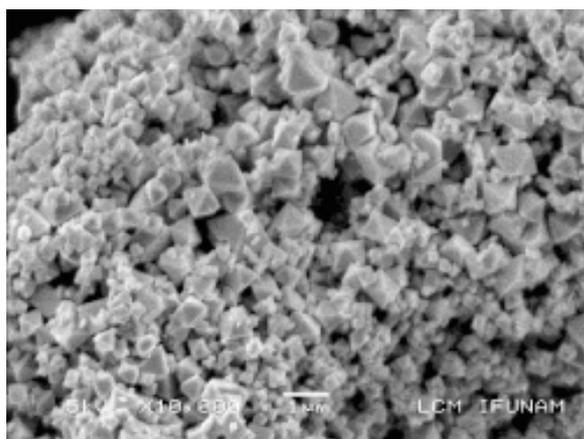
Increasing interest has been generated with antiferromagnetic nanoparticles since the discovery of their potentials for quantum tunneling [20,21] and their applications in spin-valve systems [22]. In bulk crystalline form,  $\text{Co}_3\text{O}_4$  exhibits antiferromagnetism with Néel temperature of about  $T_N = 33\text{K}$  [23].

Early studies by Néel suggested that nanoparticles of antiferromagnetic materials should exhibit superparamagnetic behavior or a weak ferromagnetism, which may be ascribed to the reduced coordination of the surface spins, leading to important changes in the magnetic order.

Corresponding author: Ana Fernandez-Osorio, e-mail: ana8485@servidor.unam.mx



**Fig. 1.** XRD pattern of  $\text{Co}_3\text{O}_4$  nanoparticles. All peaks can be indexed to  $\text{Co}_3\text{O}_4$  cobaltite, with structure given by JCPDS No.421467.



**Fig. 2.** SEM micrograph of  $\text{Co}_3\text{O}_4$  nanoparticles. Note the spinel octahedral morphology.

Makhlouf [24] reported magnetization and magnetic relaxation measurements in  $\text{Co}_3\text{O}_4$  particles with sizes about 20 nm, he observed a narrow cusp at about 25K in zero field-cooling (ZFC) magnetization and irreversibility in the field-cooling mode (FC). Both FC and ZFC modes bifurcate at lower temperatures. Above 60K magnetization temperature measurements,  $M-T$  obey Curie-Weiss law with negative Weiss temperature,  $\theta$  at about 85K.

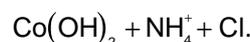
In this work, we present a facile synthesis method, in mild reaction conditions to obtain  $\text{Co}_3\text{O}_4$

nano-octahedra with average crystallite size of  $16.4 \pm 3.1$  nm, and their structural and magnetic study

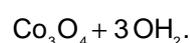
## 2. EXPERIMENTAL

### 2.1 Synthesis

In a typical synthesis, an aqueous solution of 0.4M  $\text{CoCl}_2 \cdot 6\text{H}_2\text{O}$  (Aldrich 99%) was prepared adding drop to drop 0.5 ml of 0.5 M ammonium hydroxide up to obtain a pink precipitate (pH  $8 \pm 0.5$ ) of  $\text{Co}(\text{OH})_2$  according to following reaction:

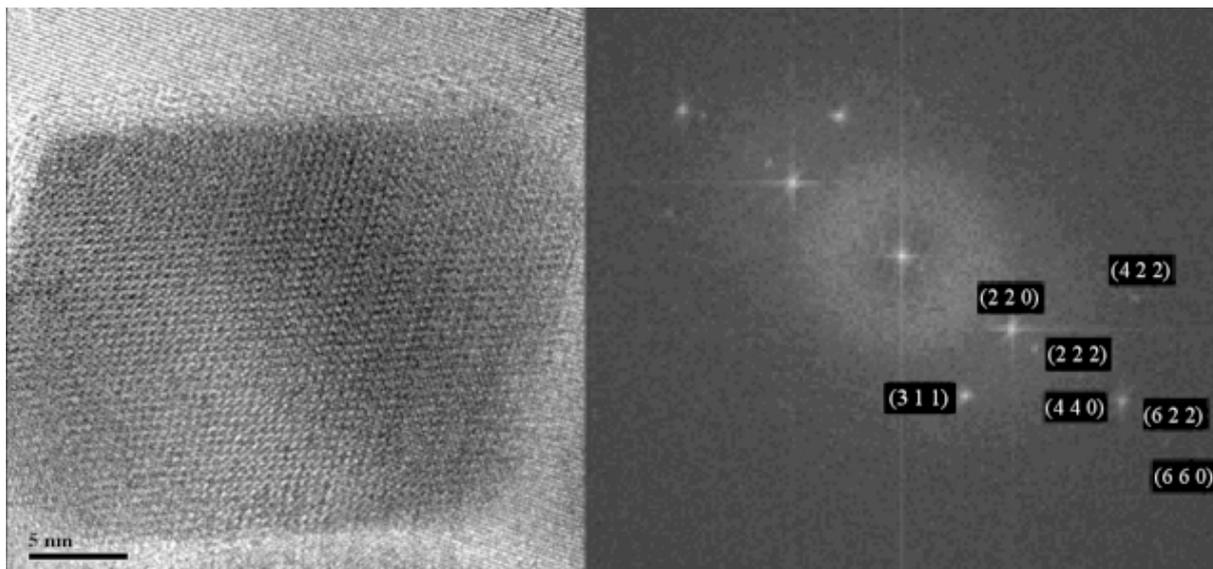


The precipitate was washed with distilled water to remove  $\text{Cl}^-$  and  $\text{NH}_4^+$  ions, and the final product was dried at room temperature and then, calcined in air at 150 °C for 2 h as follows:

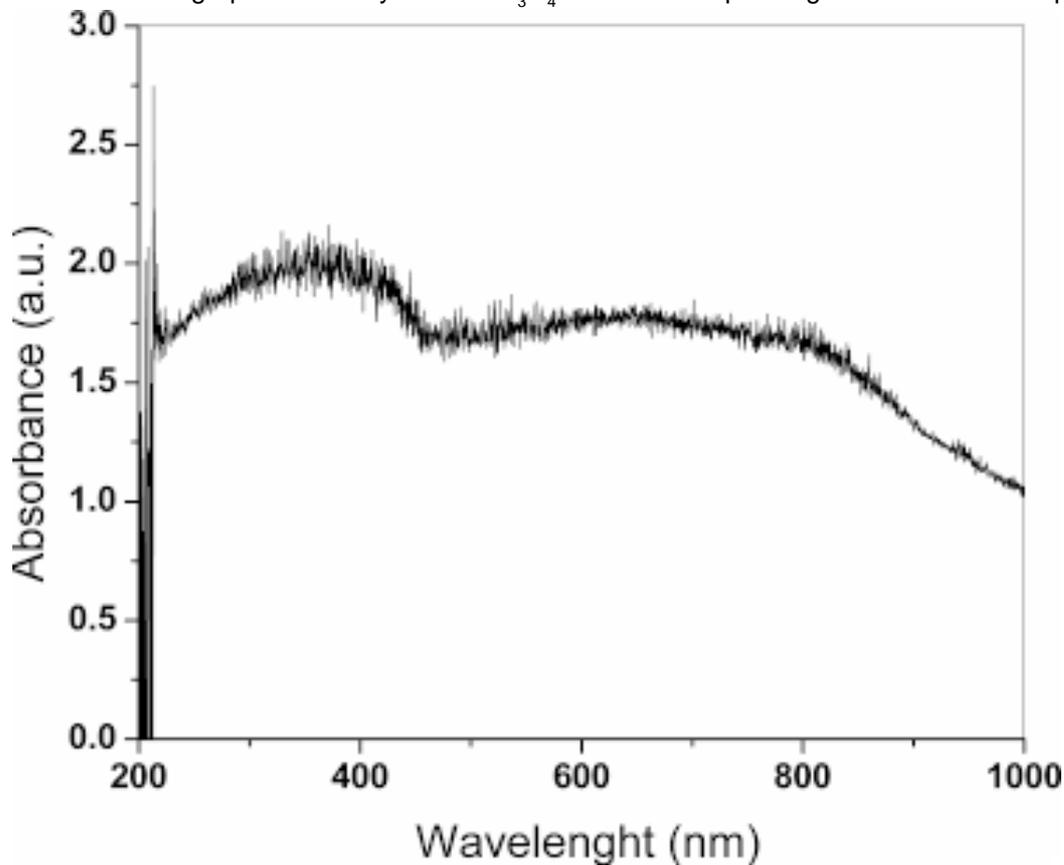


### 2.2. Characterization Techniques

X-ray diffraction pattern was obtained at room temperature with Cu  $K_\alpha$  radiation ( $\lambda = 1.5406 \text{ \AA}$ ) be-



**Fig. 3.** HR-TEM micrograph of nanocrystalline  $\text{Co}_3\text{O}_4$  and its corresponding electron diffraction pattern.



**Fig. 4.** UV-visible electronic absorption spectrum of  $\text{Co}_3\text{O}_4$  Nps.

tween  $2.5^\circ$  and  $70^\circ$  with a  $2\theta$  step of  $0.02^\circ$  for 0.8 s per point, using a D5000 Siemens diffractometer. UV-Vis electronic absorption spectrum was measured in diffuse reflectance mode in the 200–1200 nm wavelength range with an Ocean-optics

HR4000 spectrometer. Raman spectrum was obtained from 200 to  $900\text{ cm}^{-1}$  with a Nicolet Almega XR dispersive Raman spectrometer, using a scan time of 25 s and resolution of  $\sim 4\text{ cm}^{-1}$ . An Nd:YVO<sub>4</sub> 532-nm laser was used for excitation and the inci-

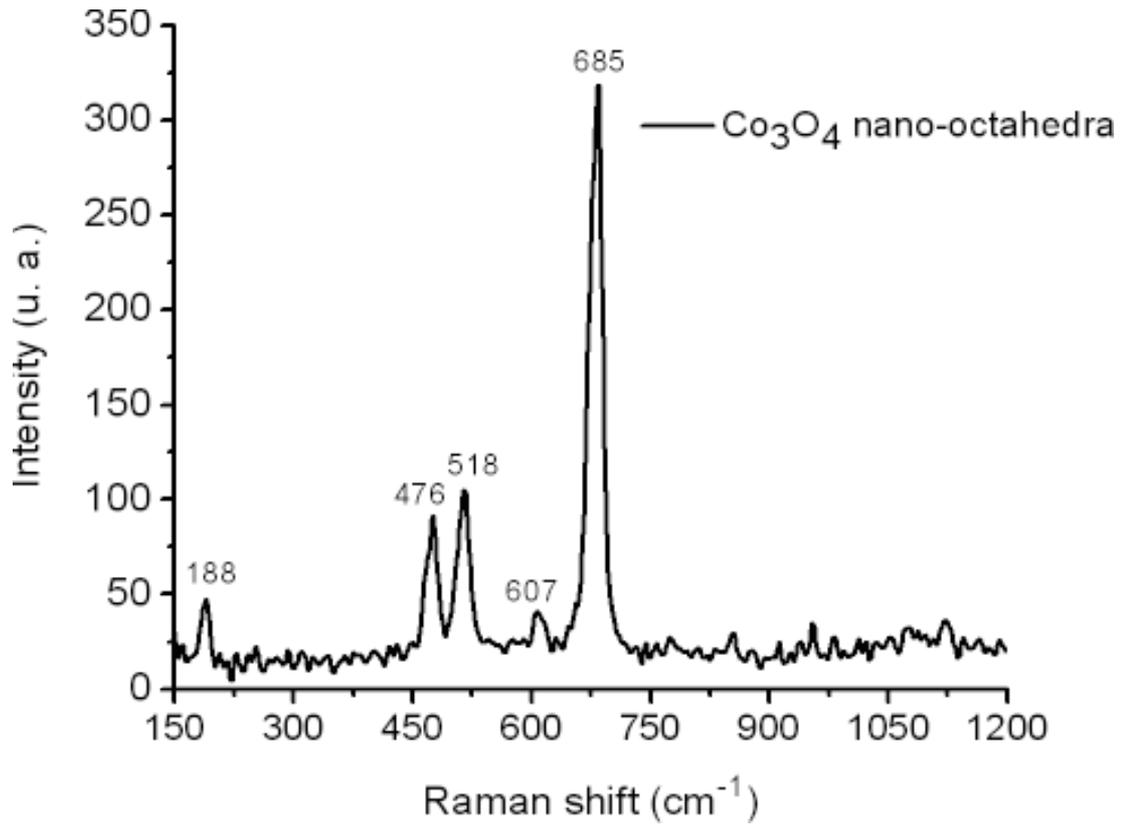


Fig. 5. Raman spectrum of  $\text{Co}_3\text{O}_4$  nano-octahedral.

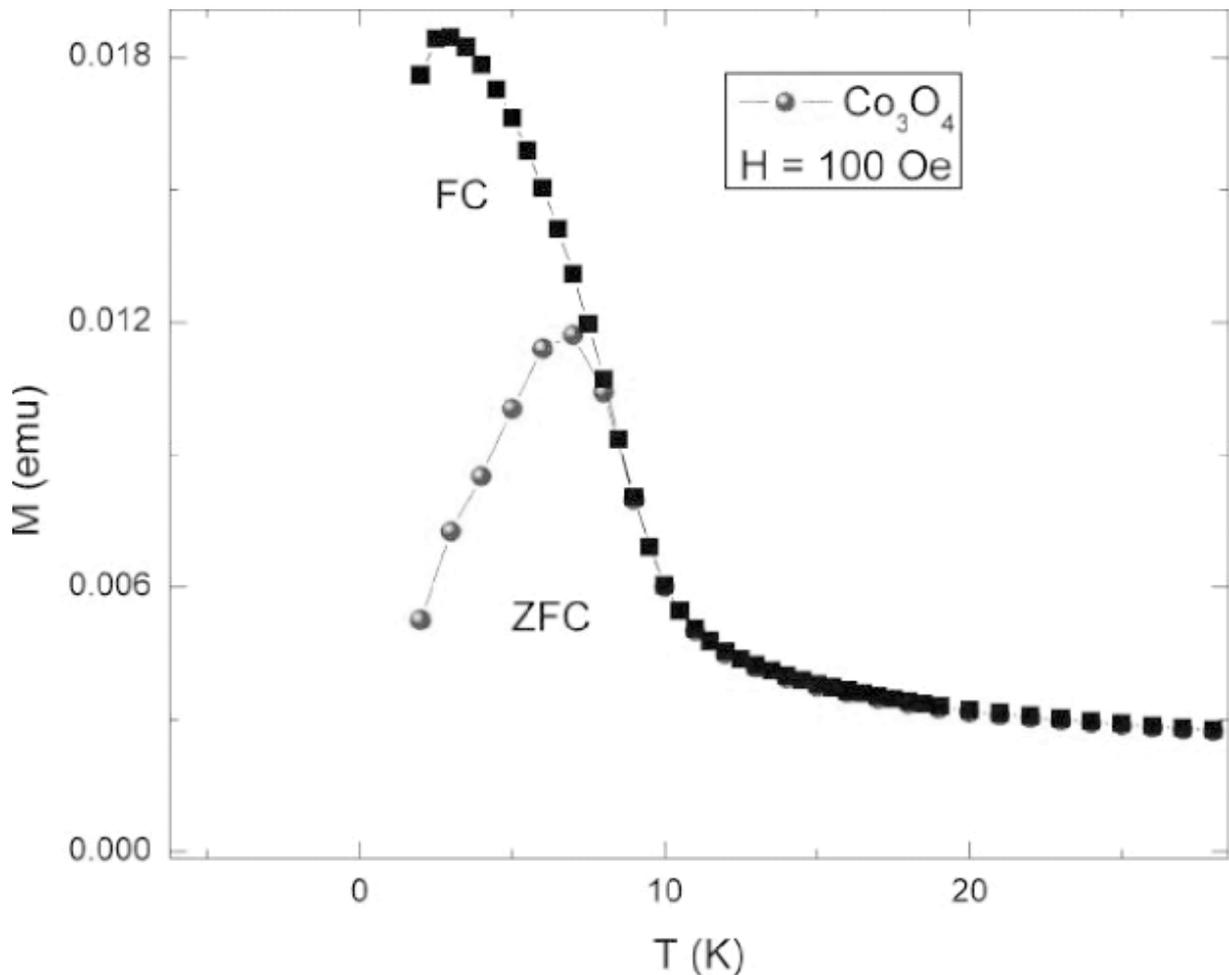


Fig. 6. Temperature dependence of FC and ZFC magnetization for  $\text{Co}_3\text{O}_4$  nanoparticles.

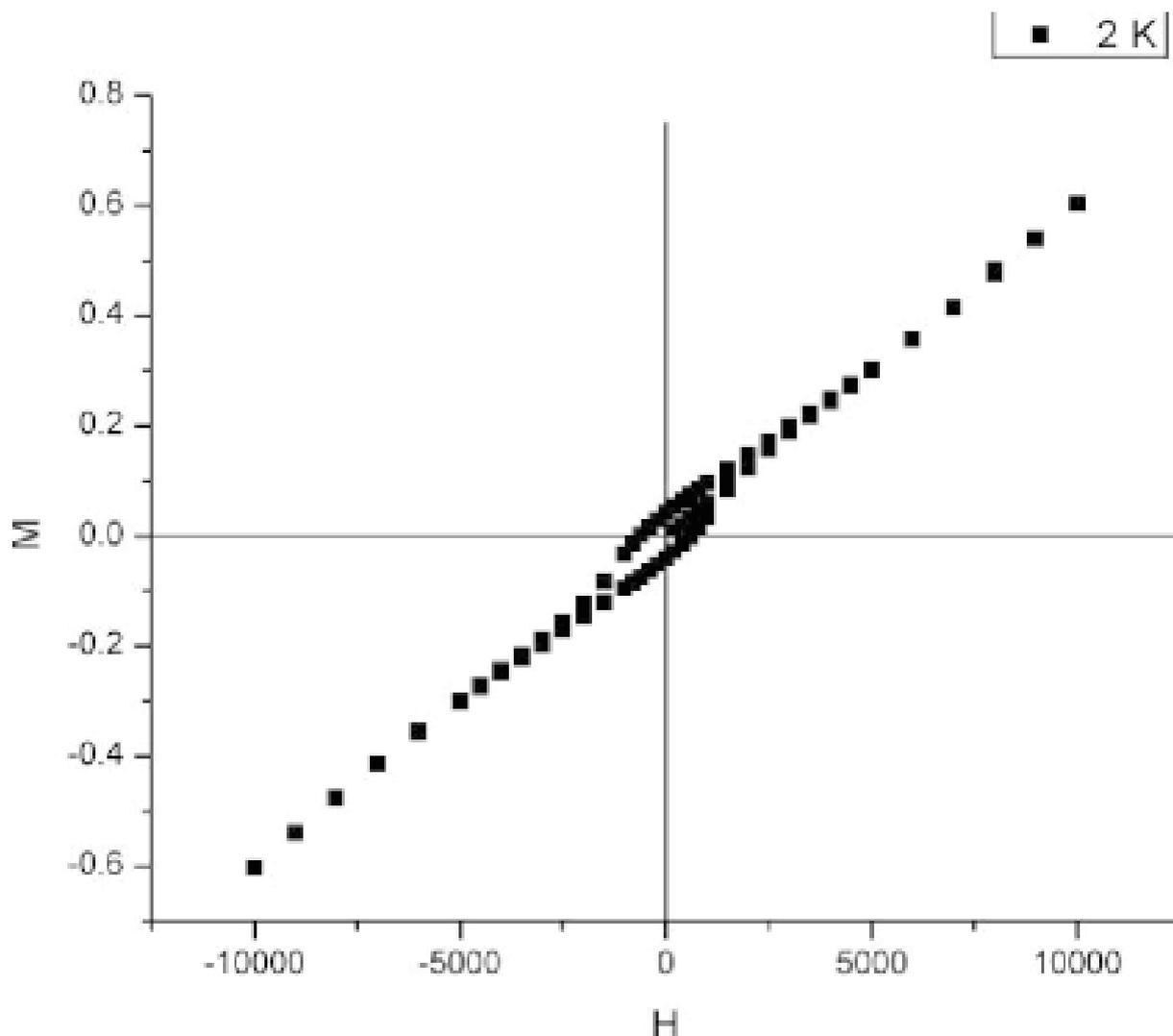


Fig. 7. Hysteresis loop obtained at 2K.

dent power on the sample was  $\sim 10$  mW. High-resolution transmission electron micrographs (HR-TEM) were obtained in a JEOL 2010 FASTEM analytical microscope operating at 200 kV, by deposition of  $\text{Co}_3\text{O}_4$  powder dispersed in methanol onto a 200-mesh Cu grid coated with carbon layer. Scanning electron micrographs (SEM) were obtained in a JEOL JSM5900 LV microscope by direct immersion of a grid into  $\text{Co}_3\text{O}_4$  powder, without the use of any solvent. Magnetic studies were performed on a MPMS SQUID Quantum Design Magnetometer on powdered sample of  $\text{Co}_3\text{O}_4$  nano-octahedra. The temperature was varied between 2 and 300K according to a zero field cooling (ZFC)/field cooling (FC) procedure at 100 Oe, and the hysteresis loop was obtained at 2K, in a magnetic field of up to  $\pm 3$ T.

### 3. RESULTS AND DISCUSSION

The X-ray powder diffraction pattern of the nanocrystalline product is showed in Fig. 1. All diffraction peaks can be perfectly indexed to cobalt oxide  $\text{Co}_3\text{O}_4$  spinel structure (JCPDS 42-1467) with a unit symmetry described by the space group  $Fd3m$  and lattice parameter  $a = 8.083$  Å. The XRD pattern reveals the high purity of the sample.

To determine the average crystallite size ( $16.4 \pm 3.1$  nm) we used the classical Scherrer equation over all reflections. The  $\text{Co}_3\text{O}_4$  nano-crystals morphology was examined by SEM. Fig. 2 shows the formation of homogeneous nano-octahedra.

The HR-TEM micrograph (Fig. 3) shows an isolated  $\text{Co}_3\text{O}_4$  nanocrystallite with dimensions of about 20.5 nm. The interplanar distances deter-

mined from their corresponding electron diffraction patterns confirm that the nanocrystals are composed of  $\text{Co}_3\text{O}_4$ .

With the aim of to study the optical response of the  $\text{Co}_3\text{O}_4$  nano-octahedra, UV-Visible electronic absorption spectroscopy using diffuse reflectance technique (DRS) was obtained. Fig. 4 shows a typical absorption spectrum of  $\text{Co}_3\text{O}_4$  nanoparticles where two wide absorption bands are observed. The first band from 250 to 450 nm involves the charge transfer transitions  $\text{O}^{2-} \rightarrow \text{Co}^{2+}$  and  $\text{O}^{2-} \rightarrow \text{Co}^{3+}$  and Co(III) in an octahedral site:  ${}^1\text{A}_{1g} \rightarrow {}^1\text{T}_{2g}$ . The second band centered about 650 nm is assigned to Co(III) in an octahedral site:  ${}^1\text{T}_{1g} \leftarrow {}^1\text{A}_{1g}$  and Co(II) in a tetrahedral site:  ${}^4\text{T}_1 \leftarrow {}^4\text{A}_2$  [25].

Furthermore, it is well known that Raman spectroscopy is a nondestructive technique which in the last years has been extensively used in nanostructure characterization. Fig. 5 shows the Raman spectrum of the nanocrystalline  $\text{Co}_3\text{O}_4$ , five active Raman modes characteristic of this cobalt oxides are evident at 188, 476, 518, 607, and 685  $\text{cm}^{-1}$ , in agreement with those reported by Hadjiev et al. [26].

### 3.2 Magnetic measurements

Magnetic  $M$ - $T$  measurements were performed at temperatures from 2 to 300K under a 100 Oe field. Fig. 6 shows the  $M$ - $T$  curves of both FC and ZFC for the  $\text{Co}_3\text{O}_4$  nano-ctahedra sample.  $T_N$  could not be observed at around 33K; however, the FC and ZFC curves were strongly bifurcated at 8K. This bifurcation temperature of FC and ZFC was defined as the blocking  $T_b$  temperature. These particles could be considered to form a single domain ordered antiferromagnetically, below  $T_N = 33\text{K}$ .

Magnetization far above  $T_b$  presents a paramagnetic behavior, whereas near above  $T_b$ , a superparamagnetic response was observed. The hysteresis curve at a temperature of 2K is given in Fig. 7. Coercivity was observed at 1200 Oe.

Below  $T_b$ , a slight hysteresis loop appears indicating a ferromagnetic behavior. This magnetic response could be explained by uncompensated surface spins of the  $\text{Co}_3\text{O}_4$  nanoparticles. Ichiyanagi et al. [27] reported similar magnetic behavior on  $\text{Co}_3\text{O}_4$  nanoparticles with an average size of between 3.1 and 9.2 nm.

### 4. CONCLUSIONS

$\text{Co}_3\text{O}_4$  nano-octahedra in a single spinel phase were obtained by a facile hydrothermal method at mild

reaction conditions. A weak ferrimagnetism is observed at temperatures below 8K due to uncompensated surface spins behaving superparamagnetically, the total magnetic spins became easy to order. At high magnetic field,  $M$ - $T$  follows a Curie-Weiss law without any irreversibility.

### ACKNOWLEDGMENTS

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