

# Application of linear and nonlinear fracture mechanics criteria for crack propagation analysis in single crystal bodies

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## Abstract

The actual problem of the crack growth prediction in anisotropic elastic and inelastic materials has not been fully investigated neither theoretically nor experimentally. In particular the analysis of crack growth in single-crystal gas-turbine blades is of great interest to assess their strength and durability. The present research is devoted to the comparison of various linear and nonlinear fracture mechanics criteria of the crack propagation in anisotropic materials under different loading conditions (fatigue, thermal fatigue and creep) with published experimental data.

The computational features of the stress intensity factors (SIF)  $K_I$ ,  $K_{II}$ ,  $K_{III}$ , resolved shear SIF  $K_{r_{ss}}$ , J-integral and  $C^*$ -integral for anisotropic materials in the framework of the finite-element analysis is considered. Verification of implemented procedures for fracture mechanics parameters is carried out on tension specimens of different crystallographic orientations. The results of the edge crack in the single crystal blade of gas-turbine engine simulations are presented and discussed.

## 1 Introduction

Single crystal nickel-based superalloys are commonly used in parts of gas turbine engines (for example 1<sup>st</sup> stage high-pressure turbine blades) that are subject to high temperatures and require high strength, excellent high temperature creep and fatigue resistance. The main difficulty during modeling of deformation process of such single crystal materials is a strong anisotropy of elastic and viscoplastic properties [1]. In addition, the prediction of the fracture of single crystal materials has been not fully investigated neither theoretically nor experimentally. There are examples of experimentally approved static strength criteria [2] and crack initiation criteria [3]. However an investigation of the crack propagation process in the monocrystalline bodies with extensive plastic zones under the complex loading is only beginning.

## 2 Stress intensity factor for anisotropic bodies

Below we use the formalism proposed by Lechnitskii [4]. Direct calculations [5] or calculations with use of the M-integral [6] are used for the stress intensity factor (SIF) evaluation for anisotropic bodies. The complexity of direct SIF computations in the anisotropic case is caused by a mutual coupling of mode I (opening, tensile) and mode II (sliding, in-plane

shear) of a crack [7]:

$$\begin{Bmatrix} K_I \\ K_{II} \\ K_{III} \end{Bmatrix} = [B]^{-1} \begin{Bmatrix} u \\ v \\ w \end{Bmatrix} \sqrt{\frac{\pi}{2r}}, \quad (1)$$

where  $u$ ,  $v$ ,  $w$  are components of displacement vector along crack faces perpendicular to front, along perpendicular to crack faces and along crack front correspondingly,  $r$  is the distance from the crack front. The matrix  $[B]^{-1}$  is introduced by the expression [7]:

$$[B]^{-1} = \begin{bmatrix} \operatorname{Re} \left[ \frac{i}{\mu_1 - \mu_2} (q_2 - q_1) \right] \frac{1}{D} & \operatorname{Re} \left[ \frac{-i}{\mu_1 - \mu_2} (p_2 - p_1) \right] \frac{1}{D} & 0 \\ \operatorname{Re} \left[ \frac{-i}{\mu_1 - \mu_2} (\mu_1 q_2 - \mu_2 q_1) \right] \frac{1}{D} & \operatorname{Re} \left[ \frac{i}{\mu_1 - \mu_2} (\mu_1 p_2 - \mu_2 p_1) \right] \frac{1}{D} & 0 \\ 0 & 0 & \sqrt{c_{44}c_{55} - c_{45}^2} \end{bmatrix}, \quad (2)$$

$$D = \begin{vmatrix} \operatorname{Re} \left[ \frac{i}{\mu_1 - \mu_2} (\mu_1 p_2 - \mu_2 p_1) \right] & \operatorname{Re} \left[ \frac{i}{\mu_1 - \mu_2} (p_2 - p_1) \right] \\ \operatorname{Re} \left[ \frac{i}{\mu_1 - \mu_2} (\mu_1 q_2 - \mu_2 q_1) \right] & \operatorname{Re} \left[ \frac{i}{\mu_1 - \mu_2} (q_2 - q_1) \right] \end{vmatrix}, \quad (3)$$

where  $\mu_1, \mu_2$  are complex roots with the positive real part of the characteristic equation

$$a_{11}\mu^4 - 2a_{16}\mu^3 + (2a_{12} + a_{66})\mu^2 - 2a_{26}\mu + a_{22} = 0, \quad (4)$$

where  $a_{ij}, c_{ij}$  are elements of the compliance and stiffness matrixes correspondingly,  $p_k$  and  $q_k$  are defined by expressions:

$$\begin{aligned} p_k &= a_{11}\mu_k^2 + a_{12} - a_{16}\mu_k, \\ q_k &= a_{12}\mu_k + \frac{a_{22}}{\mu_k} - a_{26}. \end{aligned} \quad (5)$$

However, several authors [9, 10, 11] point out that SIFs  $K_I$ ,  $K_{II}$ ,  $K_{III}$  and their traditional combinations do not adequately describe the fracture process in single crystals. Therefore the resolved shear SIF  $K_{r_{ss}}$ , taking into account the crystal slip system, was proposed in [9]:

$$K_{r_{ss}} = \lim_{r \rightarrow 0} \tau_{r_{ss}} \sqrt{2\pi r}, \quad (6)$$

where  $\tau_{r_{ss}}$  is the maximum shear stresses in the crystal slip system:

$$\tau_{r_{ss}} = \mathbf{n} \cdot \boldsymbol{\sigma} \cdot \mathbf{l}, \quad (7)$$

where  $\boldsymbol{\sigma}$  is Cauchy stress tensor,  $\mathbf{n}$  is the normal to the slip plane,  $\mathbf{l}$  is the slip direction, which can be expressed using Burgers vector  $\mathbf{b}$  as:  $\mathbf{l} = \mathbf{b}/|\mathbf{b}|$ .

According to experimental data [11, 12, 13] for face-centered single crystal nickel-based superalloys the crack propagation takes place along  $\{111\}$  planes.

### 3 Invariant contour integrals

The invariant contour integrals or else path-independent integrals [14, 15], which are obtained as a consequence of the energy balance law, can be used for the solving linear and nonlinear fracture mechanics problems. The J-integral should be used instead of SIF, if

a body with crack has extensive plastic zones. The J-integral is defined for an arbitrarily oriented crack (Fig. 1) in two-dimensional (2D) case in the absence of any body forces as:

$$\mathbf{J} = J_1 \mathbf{e}_1 + J_2 \mathbf{e}_2 = \int_C (W \mathbf{n} + \mathbf{n} \cdot \boldsymbol{\sigma} \cdot \mathbf{u} \nabla) dl, \quad (8)$$

where  $\mathbf{e}_1, \mathbf{e}_2$  are unit vectors of the coordinate system associated with the crack tip,  $\mathbf{n}$  is external normal to the integration contour C,  $\mathbf{u}$  - displacement vector. The rigorously correct form of J-integral components is:

$$J_k = \lim_{\varepsilon \rightarrow 0} \int_{C_\varepsilon} \left( W n_k + n_i \sigma_{ij} \frac{\partial u_j}{\partial x_k} \right) dl. \quad (9)$$

In equations (8) and (9) W is the strain-energy density defined as the stress-work through the mechanical strains (without thermal part):

$$W = \int_0^\varepsilon \boldsymbol{\sigma} \cdot \cdot d\boldsymbol{\varepsilon} = \int_0^t \boldsymbol{\sigma} \cdot \cdot \dot{\boldsymbol{\varepsilon}} dt. \quad (10)$$

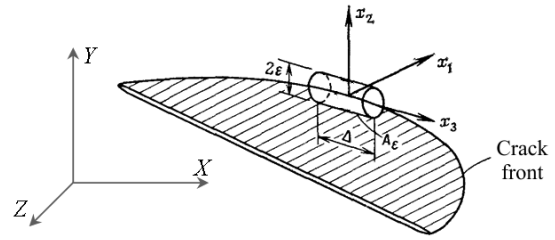
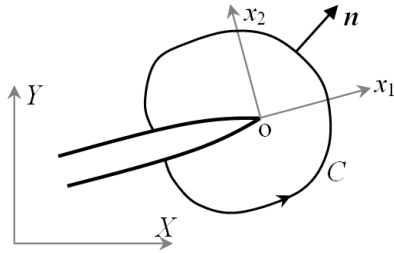


Figure 1: Crack tip and coordinate systems (global (X,Y) and crack system  $(x_1, x_2)$ ). 2D case.

Figure 2: Crack front and coordinate systems (global (X,Y,Z) and crack system  $(x_1, x_2, x_3)$ ). 3D case.

The generalization of the equation (8) for the 3D case is:

$$\mathbf{J} = J_1 \mathbf{e}_1 + J_2 \mathbf{e}_2 + J_3 \mathbf{e}_3 = \frac{1}{\Delta} \int_A (W \mathbf{n} + \mathbf{n} \cdot \boldsymbol{\sigma} \cdot \mathbf{u} \nabla) dA + J_3 \mathbf{e}_3, \quad (11)$$

where  $\Delta$  is a length of the tube along crack front (see Fig. 2). Components of the J-integral in 3D case is defined by expressions:

$$\begin{aligned} J_k &= \frac{1}{\Delta} \lim_{\varepsilon \rightarrow 0} \int_{A_\varepsilon} \left( W n_k + n_i \sigma_{ij} \frac{\partial u_j}{\partial x_k} \right) dA, \quad k = 1, 2 \\ J_3 &= \frac{1}{\Delta} \lim_{\varepsilon \rightarrow 0} \int_{A_\varepsilon} \left( W_3 n_1 + n_i \sigma_{i3} \frac{\partial u_3}{\partial x_1} \right) dA, \end{aligned} \quad (12)$$

where  $W_3$  for mode III is:

$$W_3 = \int_0^{\varepsilon_{3j}} \sigma_{3j} \cdot \cdot d\varepsilon_{3j} = \int_0^t \sigma_{3j} \cdot \cdot \dot{\varepsilon}_{3j} dt. \quad (13)$$

Note, that the components of vectors and tensors in index form correspond to the crack coordinate system  $(x_1, x_2, x_3)$ .

The invariant contour integral for a viscoelastic material behavior case [16] should be used in the presence of extensive viscous (creep) zones. Such integral can be introduced by analogy with the equation (11):

$$\mathbf{C}(t) = C_1(t)\mathbf{e}_1 + C_2(t)\mathbf{e}_2 + C_3(t)\mathbf{e}_3 = \frac{1}{\Delta} \int_A (P\mathbf{n} + \mathbf{n} \cdot \boldsymbol{\sigma} \cdot \dot{\mathbf{u}}\nabla) dA + C_3(t)\mathbf{e}_3, \quad (14)$$

$$\begin{aligned} C_k(t) &= \frac{1}{\Delta} \lim_{\varepsilon \rightarrow 0} \int_{A_\varepsilon} \left( Pn_k + n_i \sigma_{ij} \frac{\partial \dot{u}_j}{\partial x_k} \right) dA, \quad k = 1, 2, \\ C_3(t) &= \frac{1}{\Delta} \lim_{\varepsilon \rightarrow 0} \int_{A_\varepsilon} \left( P_3 n_1 + n_i \sigma_{i3} \frac{\partial \dot{u}_3}{\partial x_1} \right) dA, \end{aligned} \quad (15)$$

$$P = \int_0^{\dot{\varepsilon}} \boldsymbol{\sigma} \cdot \cdot d\dot{\boldsymbol{\varepsilon}}, \quad P_3 = \int_0^{\dot{\varepsilon}_{3j}} \sigma_{3j} \cdot \cdot d\dot{\varepsilon}_{3j}. \quad (16)$$

During the loading of a viscoelastic material the elastic strain dominates at initial time, but over time the viscous (creep) zone is becoming extensive and the value of  $\mathbf{C}(t)$  tends asymptotically to the standard fracture mechanics parameter  $C^*$  [17].

## 4 Numerical examples

The relations for the computation of the described above fracture mechanics parameters (1), (6), (11) and (14) for anisotropic materials have been implemented in the finite element program PANTOCRATOR [18]. The parameters SIFs and  $K_{r,ss}$  are calculated numerically using the method based on an extrapolation of displacements and stresses into the crack tip. The path-independent integrals J and C(t) are calculated using the equivalent domain integration (EDI) method [19].

As an illustration the problem of an edge crack in the anisotropic specimen [13] is considered below. The finite element (FE) mesh of the specimen is shown in Fig. 3. The FE model includes 4770 twenty-node isoparametric brick elements with second order of shape functions.

According to the experimental program [13] the FE computations for three different crystallographic orientations (Fig. 3) of the specimens were carried out:

- orientation A:  $\langle 00\bar{1} \rangle$  parallel to tensile axis (axis of specimen),  $[\bar{1}\bar{1}0]$  perpendicular to the crack front;
- $\pi\bar{1}0$  orientation B:  $\langle 001 \rangle$  parallel to tensile axis (axis of specimen),  $[0\bar{1}0]$  perpendicular to the crack front;
- orientation C:  $\langle \bar{1}11 \rangle$  parallel to tensile axis (axis of specimen),  $[\bar{1}\bar{1}0]$  perpendicular to the crack front.

In FE computations the loading is applied as uniform pressure on upper and lower specimen faces. The value of pressure corresponds to  $K_I = 4MPa\sqrt{m}$  for the equivalent isotropic specimen. The specimens are made from the single crystal alloy UDIMET 720 [13].

The typical von Mises stress field distribution in the specimen is presented in Fig. 4 for loading case A. Calculated SIFs along crack front are shown in Fig. 5. The crystallographic orientation of specimens has significant influence on values of SIFs. Comparison

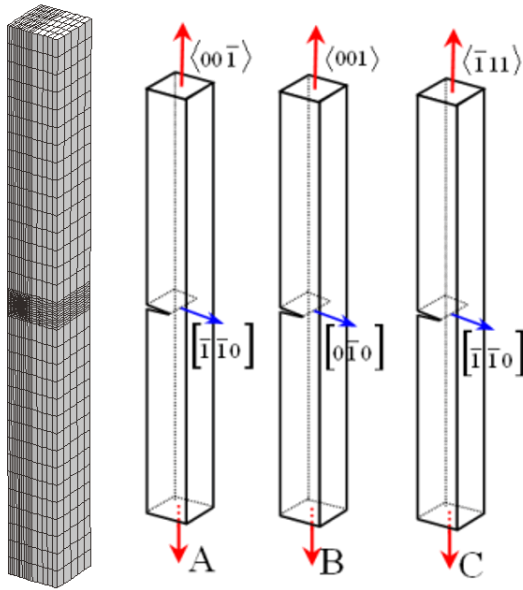


Figure 3: FE model and orientations of the cracked specimens.

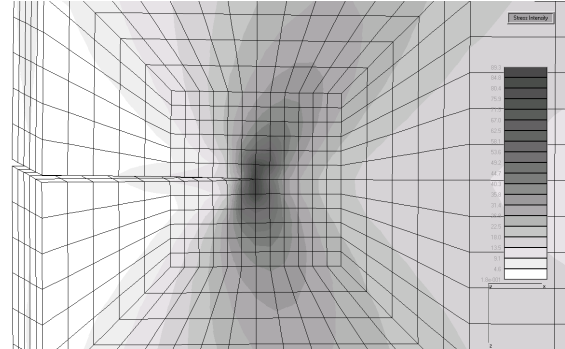


Figure 4: Von Mises stress field in the crack tip neighborhood.

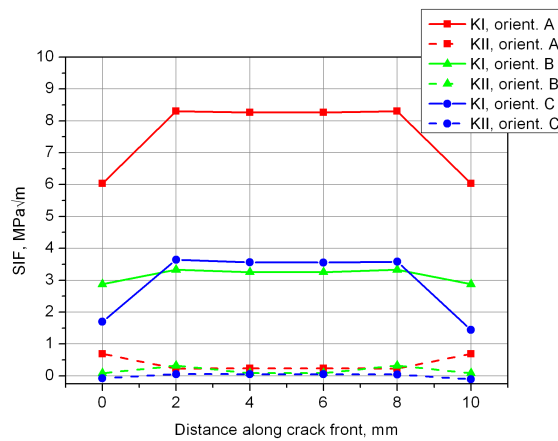


Figure 5: SIFs along crack front.

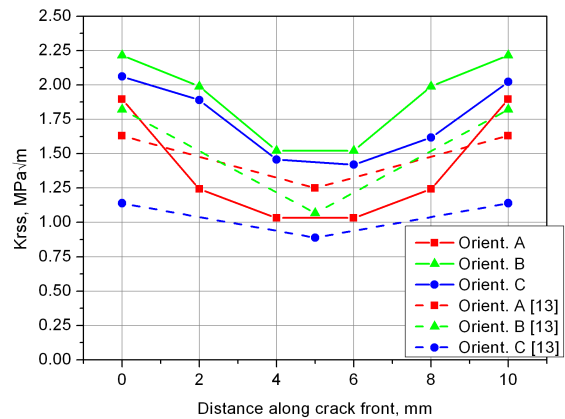


Figure 6:  $K_{rss}$  along crack front.

of calculated and experimental [13] distributions of the  $K_{rss}$  along crack front shows satisfactory agreement (Fig. 6). The computed values of J-integrals demonstrate invariance to the radius of integration contour

In another example the tensile creep of the viscoelastic plate with an edge crack is considered. The Norton creep model is used. Comparison of simulation results with the simplified analytical approximation  $C(t) = C^*(t_T/t + 1)$  is shown in Fig. 7. There is a good agreement for the times corresponding to the extended creep area.

The last example is concerned with the single crystal gas-turbine blade with an edge crack (Fig. 8). The two crack face orientations (001) and (111) are considered and compared for the blade orientation [001]. Distributions of  $J_1$ -integral and  $C_1$ -integral along crack front are shown in Fig. 9 and 10 for the crack face orientation (001). The distributions indicate that the progressive growth of fatigue and creep cracks takes place in the middle of the crack front.

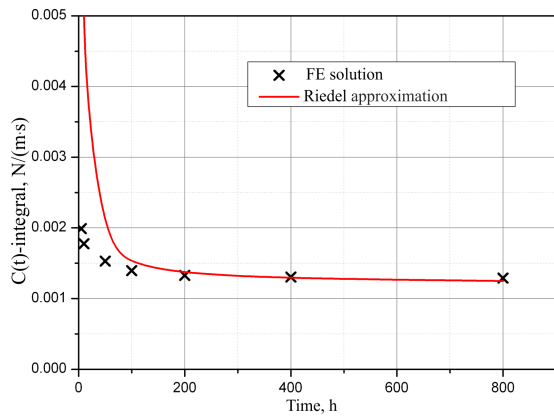


Figure 7:  $C(t)$ -integral evolution for the plate with an edge crack.

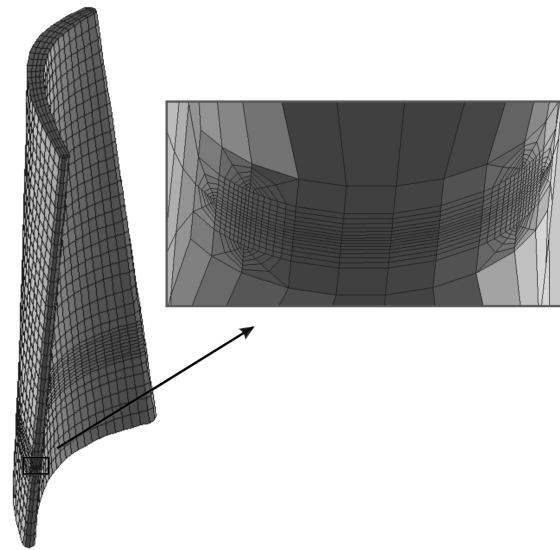


Figure 8: FE model of the single crystal turbine blade with an edge crack.

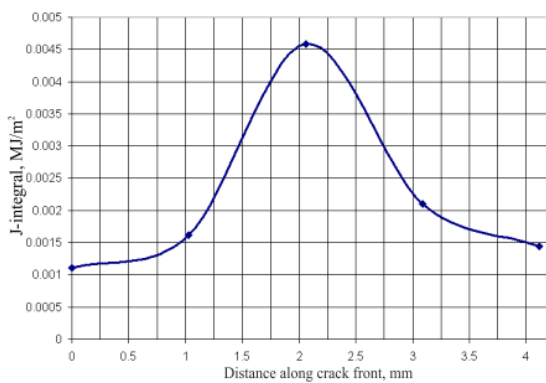


Figure 9:  $J_1$ -integral along crack front.

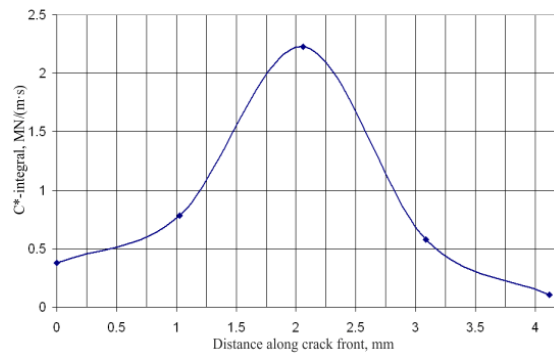


Figure 10:  $C_1^*$ -integral along crack front.

## Conclusions

A wide range of linear and nonlinear fracture mechanics methods for anisotropic materials has been considered, implemented in the FE program PANTOCRATOR and compared in simulations. Verification of implemented procedures is carried out on tension specimens of different crystallographic orientations. The further experimental studies are required for the description of fatigue, thermal fatigue and creep cracks propagation in single crystals.

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